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# Electronic structures of $\text{ReS}_2$ , $\text{ReSe}_2$ and $\text{TcS}_2$ in the real and the hypothetical undistorted structures

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Received 24 February 1997

**Abstract.** The transition-metal dichalcogenides  $\text{ReX}_2$  ( $X = \text{S}$  or  $\text{Se}$ ) and  $\text{TcS}_2$  with a  $d^3$  electron configuration have distorted  $\text{CdCl}_2$  and  $\text{Cd}(\text{OH})_2$  structures, respectively, with the  $\text{Re}(\text{Tc})$  atoms in each layer forming parallelogram-shaped connected clusters (diamond chain). *Ab-initio* band-structure calculations were performed for  $\text{ReX}_2$  and  $\text{TcS}_2$ , and the hypothetical undistorted 1T- $\text{TcS}_2$  and 3R- $\text{ReX}_2$  structures. The calculations show that  $\text{ReS}_2$ ,  $\text{ReSe}_2$  and  $\text{TcS}_2$  are semiconductors with energy gaps of about 1.0 eV, 0.5 eV and 0.7 eV, respectively, while for the undistorted structures the Fermi level is in the partly filled band of  $d_{x^2-y^2}$  and  $d_{xy}$  orbitals of the  $t_{2g}$  manifold.

X-ray photoemission spectra for the core levels and valence band of  $\text{ReSe}_2$  and  $\text{ReS}_2$  are presented. The valence x-ray photoemission spectra showed that  $\text{ReS}_2$  is a p-type semiconductor with an energy gap of about 1.5 eV, while  $\text{ReSe}_2$  is an n-type semiconductor. The experimental results are in good agreement with the band-structure calculations.

## 1. Introduction

The Tc and Re disulphides and selenides have a layered structure and semiconducting properties like the Mo and W dichalcogenides [1,2]. However, the origin of the semiconductivity is different. For the Mo and W dichalcogenides which have a  $d^2$  electron configuration and trigonal-prismatic coordination of the metal atoms by chalcogen atoms, there is a low-lying  $d_{z^2}$  band of the metal which is completely occupied by two electrons [1,3]. For the group VII dichalcogenides which have a  $d^3$  electron configuration the origin of the semiconductivity is a distortion of a  $\text{Cd}(\text{OH})_2$ - or  $\text{CdCl}_2$ -type basic structure to a structure with metal–metal bonds in parallelogram-shaped clusters along one of the in-plane axes (diamond chain). The transition can be simply thought to take place as follows: the metal atoms form zigzag chains in which two of the three d electrons are used in metal–metal bonds, followed by a Peierls distortion along the chain, leading to four-atom parallelogram-shaped clusters. In this MO scheme all d electrons are paired.

Because of their layered structure and easy cleavage parallel to the layers, STM and AFM surface studies can easily be performed [4–6]. In contrast with the transition-metal dichalcogenides such as 1T- $\text{TiS}_2$  and 2H- $\text{MoS}_2$ , the chalcogen top layer and the subsurface Re layers of  $\text{ReX}_2$  have different geometries, and STM and AFM of these compounds may show the contributions of the top chalcogen layers and the subsurface metal atoms to the images, depending on the parameters of the measurements. Such measurements have been reported by several workers, but with conflicting results and interpretations [4–9].

Structural information on the Re and Tc dichalcogenides was based on the crystal structure determination of  $\text{ReSe}_2$  by Alcock and Kjekshus [10] and unit-cell dimensions

and structures proposed on the basis of that of  $\text{ReSe}_2$  by Wildervanck and Jellinek [2] for  $\text{ReS}_2$ ,  $\text{ReSe}_2$ ,  $\text{ReSSe}$ ,  $\text{TcS}_2$  and  $\text{TcSe}_2$ . Wildervanck and Jellinek showed that  $\text{ReS}_2$ ,  $\text{ReSe}_2$  and  $\text{ReSSe}$  have a triclinic distorted  $\text{CdCl}_2$  structure with in-plane axes twice those of the hypothetical undistorted structure, while  $\text{TcS}_2$  has a similar distorted  $\text{Cd}(\text{OH})_2$ -type structure. Recently Lamfers *et al* [11] reported the x-ray single-crystal structure determinations of  $\text{ReSe}_2$ ,  $\text{ReS}_2$ ,  $\text{ReSSe}$  and  $\text{TcS}_2$  using crystals grown by Wildervanck and Jellinek [2]. Lamfers *et al* [11] showed that the structure of  $\text{ReSe}_2$  as determined by Alcock and Kjekshus is wrong in the sense that these workers interchanged inversion centres and pseudo-inversion centres. Four-atom clusters of Re atoms are present, but the Re–Re and the Re–Se distances differ strongly from those of Alcock and Kjekshus. Unit-cell dimensions and the structure of  $\text{ReS}_2$  determined by Lamfers *et al* [11] differ from those published recently by Murray *et al* [12] in the sense that a doubled  $c$  axis was found. Diamond chains similar to those of  $\text{ReSe}_2$  are present.  $\text{TcS}_2$  has a distorted  $\text{Cd}(\text{OH})_2$ -type structure with a Tc diamond chain.

Optical measurements showed energy gaps of about 1.32 eV, 1.17 eV and 1.3 eV for  $\text{ReS}_2$ ,  $\text{ReSe}_2$  and  $\text{TcS}_2$  [1, 13–15], respectively. The compounds may have application in thin-film solar cells [16], in electrochemical solar cells [17], and as catalysts for sulphur-tolerant hydrogenation and hydrosulphurization [18, 19]. Recently their use as a polarization-sensitive detector for light was reported [20].

The electronic structures of  $\text{ReS}_2$  and  $\text{ReSe}_2$  were studied by Kertesz and Hoffmann [21] and by Whangbo and co-workers [5, 22–25] in the course of their study on the origin of the lattice distortions in transition-metal dichalcogenides with  $d^n$  electron count ( $n = 1, 2, 3$ ) using extended Hückel tight-binding band-structure calculations [26]. Kertesz and Hoffmann [21] concluded that a Jahn–Teller distortion is responsible for the particular pattern of clustering in  $\text{ReSe}_2$ . Whangbo and co-workers [22–25] explained the diamond chains in these  $d^3$  compounds in terms of hidden 1D Fermi surfaces of the undistorted structures.

Until now no *ab-initio* band-structure calculation and photoemission spectra were reported for the Tc and Re dichalcogenides. In this paper we describe band-structure calculations using the localized-spherical-wave (LSW) method, with the structural data from Lamfers *et al* [11] as input. In order to recognize the driving force for the distortion the band structures of the hypothetical undistorted structures (1T- $\text{TcS}_2$ , 3R- $\text{ReS}_2$  and 3R- $\text{ReSe}_2$ ) were also calculated. The chemical bonding and structural distortions are discussed in connection with the calculated band structures. X-ray photoemission spectra for the core levels and the valence bands of  $\text{ReS}_2$  and  $\text{ReSe}_2$  were measured. The calculated results are compared with the experimental data.

## 2. Structure of $\text{ReS}_2$ , $\text{ReSe}_2$ and $\text{TcS}_2$

The crystal data determined by Lamfers *et al* [11] are listed in table 1; the space group is  $P\bar{1}$ . The axes are chosen such that  $a$  and  $b$  are the in-plane axes; the chains with clusters of metal atoms are along the shortest of the two, i.e. the  $a$  axis. The unit cells of  $\text{ReSe}_2$  and  $\text{TcS}_2$  contain one sandwich;  $\text{ReS}_2$  with a doubled  $c$  axis contains two sandwiches related by symmetry centres. For the discussion of the structural characteristics and the band-structure calculations of  $\text{ReS}_2$  we take a structure with only one sandwich per unit cell ( $c$  halved and the centre of symmetry in the Re layers). Now we have in all three cases two crystallographically independent metal atoms (M(1) and M(2)) and four crystallographically independent X ( $X = \text{S}$  or  $\text{Se}$ ) atoms (X(1), X(2), X(3) and X(4)) for all the compounds numbered in the same way. The metal atoms form chains of connected clusters of four M atoms along the  $a$  axes as shown in figure 1. The M atoms in the clusters are related by

inversion centres; the shortest metal–metal distance is for the central bond of the cluster between the M(2) atoms related by the inversion centre at  $1/2, 1/2, 1/2$ . The M–M distances in the cluster of figure 2 are given in table 2. The M atoms are in distorted octahedra of chalcogen atoms; the six independent M–X distances differ considerably, but the averages are about the same for the two independent metal atoms (2.392 Å, 2.396 Å and 2.521 Å for  $\text{TcS}_2$ ,  $\text{ReS}_2$  and  $\text{ReSe}_2$ , respectively). The X(1) and X(2) atoms are inside the space above the rows of M atoms forming the cluster chain. X(3) and X(4) are between adjacent chains. X(1) and X(2) are at a larger distance from the plane  $z = \frac{1}{2}$  (approximately the plane of the metal atoms) than the X(3) and X(4) atoms (table 3). Each X atom is coordinated approximately trigonal-pyramidally to three M atoms. The average X–M distances vary in the sequence  $X(4) > X(3) > X(2) > X(1)$ ; those for  $\text{ReSe}_2$  are given in table 3. The X layers are much more distorted from hexagonal symmetry than found by Alcock and Kjekshus [10] for  $\text{ReSe}_2$ . The Se–Se distances along rows of Se, e.g. parallel to [100], show an alternation in length of about 10%. For the [100] direction, one row consists of alternate Se(1) and Se(2) with distances of 3.417 and 3.219 Å, and the adjacent rows of alternate Se(3) and Se(4) with distances of 3.455 and 3.163 Å. The rows deviate somewhat from linearity, e.g. the angles Se(1)–Se(2)–Se(1) and Se(3)–Se(4)–Se(3) are  $167.6^\circ$  and  $170.9^\circ$  respectively. Approximately the same structural features are found for  $\text{ReS}_2$  and  $\text{TcS}_2$ .

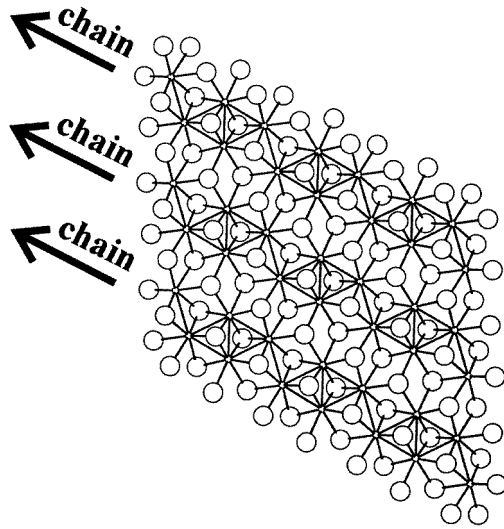
**Table 1.** Unit-cell dimensions of  $\text{TcS}_2$ ,  $\text{ReS}_2$  and  $\text{ReSe}_2$ .  $Z$  is the number of formula units per cell.  $a$  and  $b$  are the in-plane axes; the chains of metal clusters are along the  $a$  axis. The space group is  $P\bar{1}$ .

	$a$ (Å)	$b$ (Å)	$c$ (Å)	$\alpha$ (deg)	$\beta$ (deg)	$\gamma$ (deg)	$Z$
$\text{TcS}_2$	6.371	6.464	6.654	62.94	103.61	119.00	4
$\text{ReS}_2$	6.352	6.446	12.779	91.51	105.17	118.97	8
$\text{ReSe}_2$	6.597	6.710	6.721	91.84	104.90	118.91	4

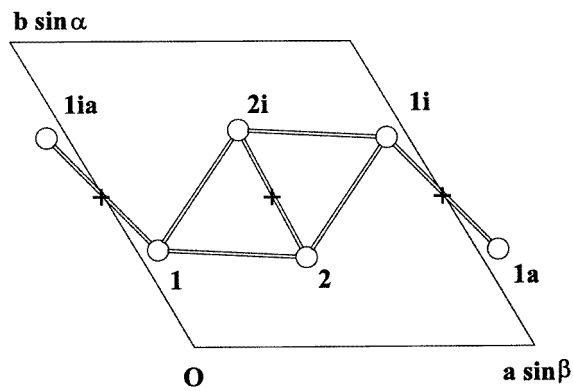
**Table 2.** Metal–metal distances in the diamond chain of  $\text{ReSe}_2$ ,  $\text{ReS}_2$  and  $\text{TcS}_2$ . The numbering of the distances refers to figure 2.

	M–M distance (Å)		
	$\text{ReSe}_2$	$\text{ReS}_2$	$\text{TcS}_2$
1–2	2.871	2.808	2.817
1– $2_i$	2.844	2.785	2.788
2– $2_i$	2.733	2.693	2.700
1– $1_{ia}$	2.987	2.881	2.8981
2– $1_a$	3.734	3.551	3.562

The hypothetical undistorted structures are of the  $\text{Cd}(\text{OH})_2$  type, space group  $P\bar{3}m1$ , for 1T- $\text{TcS}_2$  and of the  $\text{CdCl}_2$  type, space group  $R\bar{3}m$ , for  $\text{ReS}_2$  and  $\text{ReSe}_2$ . The unit-cell dimensions of the undistorted structures in table 4 are obtained with the assumption that the distortion causes a doubling of the in-plane dimensions of the unit cell of the undistorted structure and no change in the distance between neighbouring sandwiches. The ratio  $c/a$  ( $c$  taken for one sandwich) is as high as 1.90, which means that the metal atoms are in trigonal elongated octahedra of chalcogen atoms as for example in 1T- $\text{VSe}_2$ . The Re–S, Re–Se and Tc–S distances, calculated with  $z = 0.25$  for S and Se, are also listed in table 4. The metal–metal distances are equal to the length of the  $a$  axis.



**Figure 1.** The basal plane of  $\text{ReSe}_2$  with chains of Re clusters along  $[100]$  as a prototype for all three compounds. The large circles represent Se atoms, and the small circles Re atoms.



**Figure 2.** Projection of the metal cluster chain along the stacking axis  $c$ . Inversion centres are indicated by  $+$ . The distances between the numbered atoms are given in table 2. Note that the X atoms X(1) and X(2) at  $z = 3/4$  project inside the triangles  $2-2_i-1_i$  and  $1-1_{ia}-1_i$ , respectively. X(3) and X(4) project between chains.

### 3. Band-structure calculations

#### 3.1. Calculation method

*Ab initio* band-structure calculations were performed with the LSW [27] method using a scalar-relativistic Hamiltonian. We used local-density exchange–correlation potentials [28] inside space filling, and therefore overlapping spheres around the atomic constituents. The self-consistent calculations were carried out including all core electrons.

Iterations were performed with  $k$  points distributed uniformly in the irreducible part of the first Brillouin zone (BZ), corresponding to a volume of the BZ per  $k$  point of the

**Table 3.** The coordinate  $z$ , the height above the plane  $z = 1/2$  and the selenium–rhenium bond distances for Se(1), Se(2), Se(3) and Se(4) of  $\text{ReSe}_2$ .

	Se(1)		Se(2)		Se(3)		Se(4)	
$z$	0.793 86		0.761 32		0.723 66		0.696 64	
Height (Å)	1.876		1.668		1.429		1.255	
Re(2) (Å)	2.472	Re(1)	2.534	Re(1)	2.494	Re(1)	2.582	
Re(1) (Å)	2.469	Re(1)	2.433	Re(2)	2.494	Re(2)	2.574	
Re(2) (Å)	2.466	Re(2)	2.531	Re(1)	2.623	Re(2)	2.591	
Average (Å)	2.476	Average	2.496	Average	2.537	Average	2.582	

**Table 4.** Space groups, unit-cell dimensions, the ratios  $c/a$  per sandwich,  $Z$  (the number of formula units per cell) and the M–X bond distances of the hypothetical undistorted structures 1T- $\text{TcS}_2$ , 3R- $\text{ReS}_2$  and 3R- $\text{ReSe}_2$ ; the M–X distance is calculated with the  $z$  coordinate of S and Se equal to 0.25.

Compound	Space group	$a$ (Å)	$c$ (Å)	$Z$	$c/a$	M–X distance (Å)
1T- $\text{TcS}_2$	$\text{P}\bar{3}\text{m}1$	3.209	5.924	1	1.846	2.372
3R- $\text{ReS}_2$	$\text{R}\bar{3}\text{m}$	3.200	18.194	3	1.895	2.390
3R- $\text{ReSe}_2$	$\text{R}\bar{3}\text{m}$	3.327	19.018	3	1.905	2.490

**Table 5.** Input parameters of atoms (Re, Se; Tc, S) and empty spheres (Va) for the band-structure calculations of 3R- $\text{ReX}_2$  ( $X = \text{S}$  or  $\text{Se}$ ) and 1T- $\text{TcS}_2$ , and the output of the electronic configurations.  $R_{WS}$  represents the Wigner–Seitz radius.

Atom	$x$	$y$	$z$	$R_{WS}$ (Å)	Electronic configuration
1T- $\text{TcS}_2$					
Tc	1a	0	0	1.1623	$[\text{Kr}] 4d^{4.75}5s^{0.16}5p^{0.20}4f^{0.03}$
S	2d	1/3	2/3	1.6789	$[\text{Ne}] 3s^{1.91}3p^{4.56}3d^{0.29}$
Va	1b	0	1/2	1.1623	$1s^{0.11}2p^{0.15}3d^{0.09}$
3R- $\text{ReS}_2$					
Re	3a	0	0	1.1695	$[\text{Xe}] 5d^{4.42}6s^{0.21}6p^{0.20}5f^{0.03}$
S	6c	0	1/4	1.6892	$[\text{Ne}] 3s^{1.91}3p^{4.63}3d^{0.37}$
Va	3b	0	1/2	1.1695	$1s^{0.11}2p^{0.13}3d^{0.09}$
3R- $\text{ReSe}_2$					
Re	3a	0	0	1.2302	$[\text{Xe}] 5d^{4.56}6s^{0.27}6p^{0.23}5f^{0.03}$
Se	6c	0	1/4	1.7535	$[\text{Ar}] 4s^{1.91}4p^{4.13}4d^{0.52}4f^{0.22}$
Va	3b	0	1/2	1.2302	$1s^{0.11}2p^{0.13}3d^{0.08}$

order of  $3 \times 10^{-5} \text{ \AA}^{-3}$ . Self-consistency was assumed when the changes in the local partial charges in each atomic sphere decreased to the order of  $10^{-5}$ .

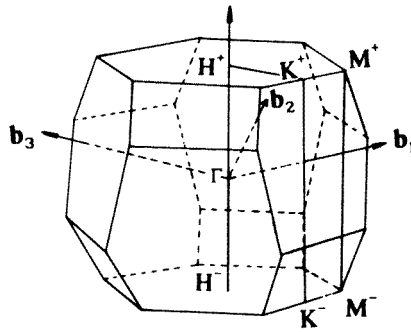
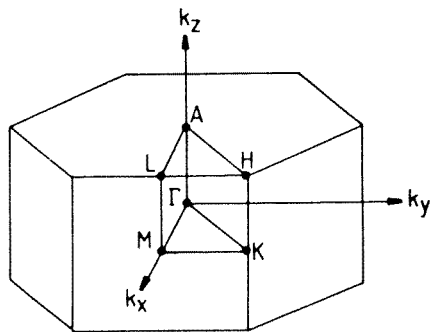
In the construction of the LSW basis [27, 29], the spherical waves were augmented by solutions of the scalar-relativistic radial equations labelled by the following atomic symbols: 3s, 3p and 3d; 4s, 4p and 4d; 5s, 5p and 4d; 6s, 6s and 5d. These correspond to the valence levels of the elements S, Se, Tc and Re, respectively. External  $l$  summation, which is used to augment a Hankel function at surrounding atoms, was extended to  $l = 3$ , resulting in the use of 4f orbitals for Se, Tc and Re. When the crystal is not very densely packed,

as in the case of the layered compounds, it is necessary to include empty spheres in the calculations. The empty spheres are located at the octahedral site in the van der Waals gaps. The functions 1s and 2p, and 3d as an extension, were used for empty spheres.

### 3.2. Electronic structure of hypothetical 1T-TcS<sub>2</sub> and 3R-ReX<sub>2</sub> (X = S or Se)

Table 5 lists the input parameters (unit-cell dimensions, atomic positions and Wigner–Seitz radii) and calculated electronic configurations of the atoms for 1T-TcS<sub>2</sub> and for 3R-ReX<sub>2</sub> (X = S or Se). We remark that not too much significance should be attributed to the orbital configuration in table 5, as these numbers depend strongly on the Wigner–Seitz radii and the presence of empty spheres. Part of the d orbital occupation is due to hybridization contribution of d orbitals in occupied s or p orbitals.

The BZs for 1T-TcS<sub>2</sub> and 3R-ReX<sub>2</sub> are shown in figures 3 and 4, respectively. Partial and total density of states (DOS), as well as the dispersion of the energy bands, are shown in figures 5–7. Energies, dominant orbital characters (OC) and symmetries at point  $\Gamma$  of the BZ are listed in table 6.

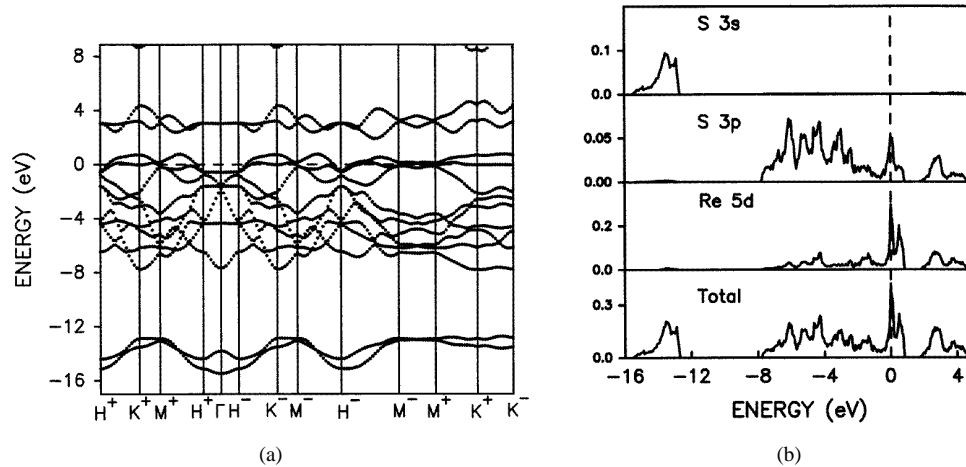


**Figure 3.** BZ and high-symmetry points for 1T-TcS<sub>2</sub>. **Figure 4.** BZ and high-symmetry points for 3R-ReX<sub>2</sub>.

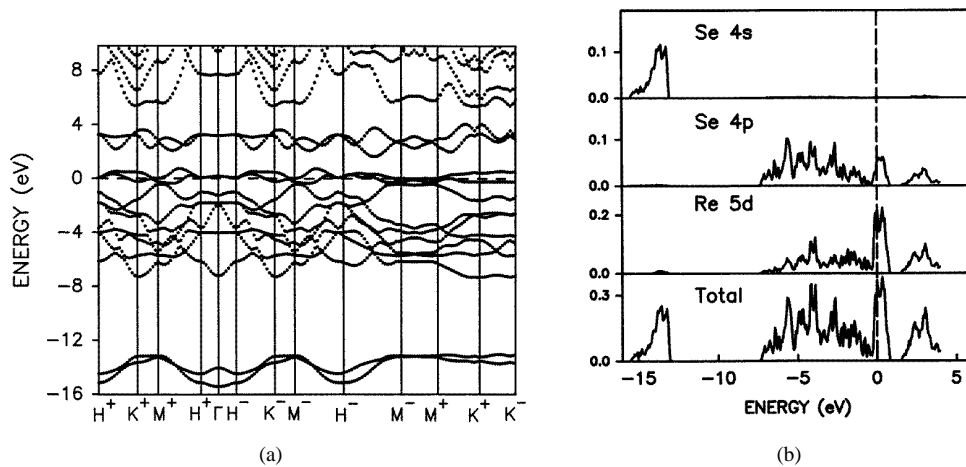
The calculated band structures of 1T-TcS<sub>2</sub> and 3R-ReX<sub>2</sub> are very similar owing to the trigonal-antiprismatic coordination of the metal atoms by chalcogen atoms. The band structures are composed of three main bands separated by energy gaps. The lowest band consist of two S 3s or Se 4s bands; the next six bands are mainly S 3p or Se 4p hybridized with Tc 4d or Re 5d. Three Tc 4d (Re 5d) orbitals form a non-bonding  $t_{2g}$  band with  $d_{z^2}$  at the bottom and  $d_{x^2-y^2}$  and  $d_{xy}$  at the top of the band. The  $d_{z^2}$  orbital is the lowest because of the large  $c/a$  ratio (note that the  $z$  axis is perpendicular to the layers). The next two bands form the antibonding  $e_g$  manifold of the transition metal. The crystal-field splitting between the  $t_{2g}$  and the  $e_g$  band is easily seen.

1T-TcS<sub>2</sub> and 3R-ReX<sub>2</sub> are metallic with three electrons in the d states of the transition-metal atoms. Two of the three electrons are in the Tc 4d (Re 5d) orbitals strongly hybridized with the chalcogen p states. The bands show a broad dispersion over all the valence bands. One d (Tc 4d or Re 5d) electron is in the two orbitals at the upper part of the valence band. These two orbitals are mainly  $d_{x^2-y^2}$  and  $d_{xy}$  of Re or Tc hybridized with some  $p_x$  and  $p_y$  orbitals of the chalcogen atoms; they have a small dispersion and are therefore quite localized (with a band width of about 1.3 eV).

According to the dispersion of the energy bands, the Fermi surfaces for the undistorted structures 1T-TcS<sub>2</sub> and 3R-ReX<sub>2</sub> can be mapped out. The structures of the Fermi surfaces



**Figure 5.** (a) Dispersion of the energy bands and (b) partial and total DOSs for 3R- $\text{ReS}_2$ .

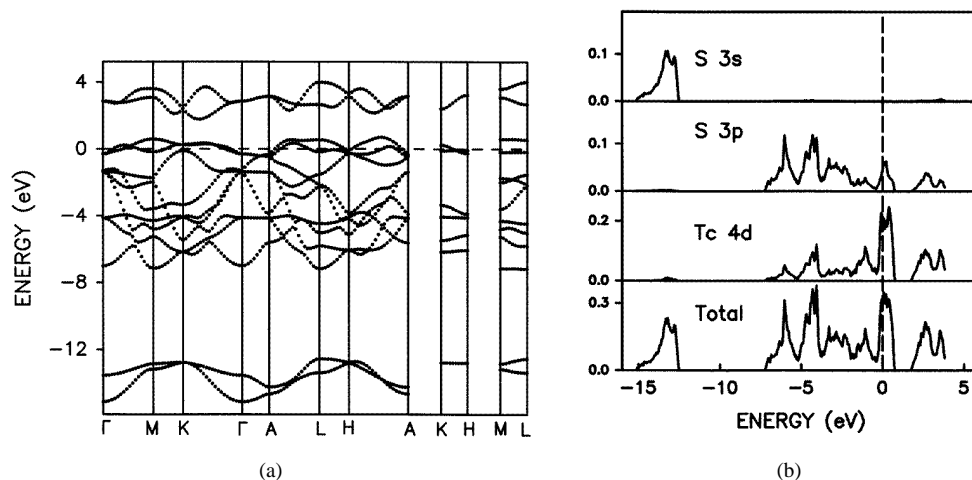


**Figure 6.** (a) Dispersion of the energy bands and (b) partial and total DOSs for 3R- $\text{ReSe}_2$ .

of the three hypothetical compounds are quite different. The Fermi surfaces for 1T- $\text{TcS}_2$  and 3R- $\text{ReS}_2$  are complex. For  $\text{TcS}_2$  there are two cylinders around  $\Gamma$ -A, six cylinders around M-L, and six ellipsoids around H, which are all occupied by electrons. For 3R- $\text{ReS}_2$  there are two cylinders along H- $\Gamma$  and 12 ellipsoids around the M points, which are occupied by electrons. For 3R- $\text{ReSe}_2$  the  $\Gamma$ -H cylinders are unoccupied.

The Fermi surfaces for the three hypothetical structure are quite three dimensional. The calculations show that S  $3p_z$  (Se  $4p_z$ ) bands have a dispersion of 2–3 eV and that the interactions in the interlayer direction are strong, although the net interaction across the van der Waals gap is weak. It is well known that Fermi surface nesting is the driving force for small distortions of the structure. Therefore 2D models such as (hidden) Fermi surface nesting leading to doubling of the  $a$  and  $b$  axes, as proposed by Whangbo and co-workers [22–25] as the origin of the distortion, is just a first-order approximation and not sufficient to describe the large distortions observed in the Re and Tc dichalcogenides.





**Figure 7.** (a) Dispersion of the energy bands and (b) partial and total DOSs for 1T-TcS<sub>2</sub>.

At the Fermi level the DOS is very high. In combination with electron–phonon interactions, this results in an instability of the undistorted crystal structure. The calculations show that two of the three d electrons are strongly hybridized with the  $p_x$  and  $p_y$  orbitals of the chalcogen atoms, and there is only one electron in the two localized  $d_{x^2-y^2}$  and  $d_{xy}$  orbitals of the transition-metal atoms. The transition is therefore expected to affect the in-plane structure of the transition-metal atoms. The hexagonal layers of S (Se) limit deformation too far from the original structure, and the  $d_{x^2-y^2}$  and  $d_{xy}$  states will remain approximately degenerate. Overlap of the  $d_{x^2-y^2}$  and  $d_{xy}$  orbitals of neighbouring metal atoms, filling the orbitals with four electrons and opening a gap, is possible when a four-atom cluster is formed. The only reasonable geometry to bring four transition atoms together on a hexagonal plane is a diamond chain.

### 3.3. Electronic structures of TcS<sub>2</sub> and ReX<sub>2</sub>

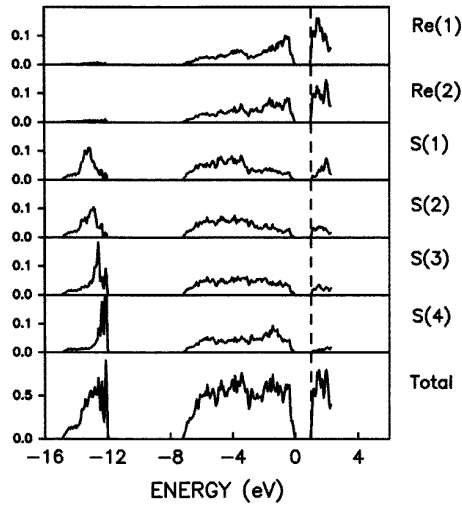
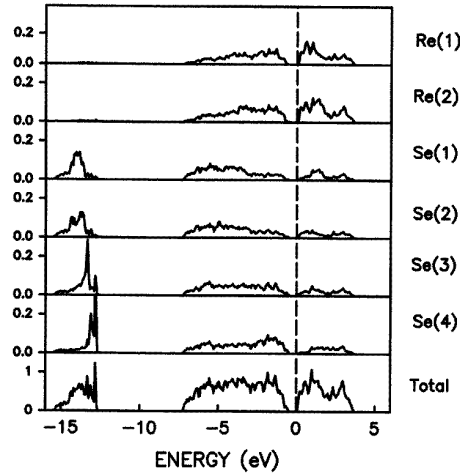
The input parameters for the band-structure calculations are given in table 7. The partial and total DOSs in figures 8–10 show that the compounds are semiconductors.

For TcS<sub>2</sub> the S 3s band ranges from  $-15.2$  to  $-12.4$  eV when the Fermi level is set at the bottom of the conduction band, compared with from  $-15.2$  to  $-12.5$  eV for 1T-TcS<sub>2</sub>. For the different S atoms the S 3s states have different characteristic peaks, because of the different coordinations by Tc atoms (similar to that for ReSe<sub>2</sub> in table 3). The valence band, consisting of S 3p hybridized with Tc 4d states, begins at  $-7.25$  eV and ends at  $-0.7$  eV. The energy gap is therefore 0.7 eV. A clear  $t_{2g}$ – $e_g$  splitting of the octahedral crystal field as seen in the undistorted structure is no longer visible. The calculations show that the conduction band is mainly composed of Tc 4d orbitals mixed with some S 3p states. The top of the valence band and the bottom of the conduction band are at the  $\Gamma$  point. The DOS of TcS<sub>2</sub> has S 3s and S 3p states similar to those of the undistorted structure because the Tc–S bonds do not change much in the two structures.

The DOSs for ReS<sub>2</sub> and ReSe<sub>2</sub> are quite similar to that of TcS<sub>2</sub> (figures 8–10). For ReSe<sub>2</sub> the Se 4s band has a width of 2.7 eV (from  $-15.4$  to  $-12.7$  eV with respect to the Fermi level at the bottom of the conduction band), similar to that of 3R-ReSe<sub>2</sub>. The valence band from  $-7.3$  to  $-0.5$  eV consists of mainly Se 4p hybridized with Re 5d states.

**Table 6.** Energies, dominant OCs and symmetry (according to Miller and Love [13]) of the states at  $\Gamma$  for 3R- $\text{ReX}_2$  and 1T- $\text{TcS}_2$   $nd^*$  indicates a mix of  $nd_{x^2-y^2}$ ,  $nd_{xy}$ ,  $nd_{xz}$  and  $nd_{yz}$  orbitals.

3R- $\text{ReS}_2$			3R- $\text{ReSe}_2$			1T- $\text{TcS}_2$		
$E$ (eV)	Symmetry	OC	$E$ (eV)	Symmetry	OC	$E$ (eV)	Symmetry	OC
-15.47	1+	S 3s	-15.48	1+	Se 4s	-15.13	1+	S 3s
-13.79	2-	S 3s	-14.06	2-	Se 4s	-13.55	2-	S 3s
-7.68	1+	S $3p_z$	-7.19	1+	Se $4p_z$	-7.00	1+	S $3p_z$
-4.35	3+	S $3p_x p_y$	-3.99	3+	Re $5d^*$	-4.12	3+	Tc $4d^*$
-1.81	2-	S $3p_z$	-1.89	1+	Re $5d_{z^2}$	-1.38	3+	S $3p_x p_y$
-1.69	1+	Re $5d_{z^2}$	-1.75	3+	Se $4p_x p_y$	-1.34	1+	Tc $4d_{z^2}$
-1.60	3+	Re $5d^*$	-1.27	2-	Se $4p_z$	-1.32	2-	S $3p_z$
-0.57	3-	S $3p_x p_y$	+0.14	3-	Se $4p_x p_y$	-0.39	3-	S $3p_x p_y$
3.03	3+	Re $5d^*$	3.12	3+	Re $5d^*$	2.85	3+	Tc $3d^*$

**Figure 8.** Partial and total DOSs for  $\text{ReS}_2$ .**Figure 9.** Partial and total DOSs for  $\text{ReSe}_2$ .

The bottom of the conduction band is composed mainly of Re  $5d$  states mixed with a small amount of Se  $4p$  orbitals.  $\text{ReSe}_2$  is therefore a semiconductor with an energy gap of about 0.5 eV. The calculations show that neither the top of the valence band nor the bottom of the conduction band is at the  $\Gamma$  point, and the direct energy gap at  $\Gamma$  is about 1.1 eV.

For  $\text{ReS}_2$  the S  $3s$  band ranges from  $-15.9$  to  $-13.0$  eV with respect to the Fermi level at the bottom of the conduction band. The valence band has a width of 7.2 eV (from  $-8.2$  to  $-1.0$  eV). There is an energy gap of about 1.0 eV between the valence and the conduction band. Kelty *et al* [5] obtained a band gap of 1.27 eV for a single  $\text{ReS}_2$  layer and 0.81 eV for the bulk from the calculations.

It is remarkable that the DOSs of the valence bands of  $\text{ReX}_2$  and  $\text{TcS}_2$  have a very broad and flat shape, in contrast with those of hypothetical 1T- $\text{TcS}_2$  and 3R- $\text{ReX}_2$ . In the real structures all the three  $d$  electrons are dispersive over the whole valence band and are strongly hybridized with S  $3p$  (Se  $4p$ ) states. This is a consequence of a stronger overlap

**Table 7.** Input parameters of atoms (Tc, Re, Se, S) and empty spheres (Va) and the output of electronic configurations.  $R_{WS}$  represents the Wigner–Seitz radius.

Atom	$x$	$y$	$z$	$R_{WS}$ (Å)	Electronic configuration
<b>TcS<sub>2</sub></b>					
Tc(1)	0.939 67	0.693 58	0.490 41	1.141	[Kr] 4d <sup>4.75</sup> 5s <sup>0.15</sup> 5p <sup>0.18</sup> 4f <sup>0.02</sup>
Tc(2)	0.489 53	0.285 05	0.502 85	1.141	[Kr] 4d <sup>4.74</sup> 5s <sup>0.15</sup> 5p <sup>0.18</sup> 4f <sup>0.02</sup>
S(1)	0.3611	0.3076	0.7900	1.648	[Ne] 3s <sup>1.90</sup> 3p <sup>4.50</sup> 3d <sup>0.29</sup>
S(2)	0.8240	0.2802	0.7622	1.623	[Ne] 3s <sup>1.89</sup> 3p <sup>4.40</sup> 3d <sup>0.23</sup>
S(3)	0.1434	0.1714	0.2741	1.699	[Ne] 3s <sup>1.92</sup> 3p <sup>4.60</sup> 3d <sup>0.30</sup>
S(4)	0.3211	0.8124	0.7011	1.759	[Ne] 3s <sup>1.93</sup> 3p <sup>4.80</sup> 3d <sup>0.33</sup>
Va(1)	0.0000	0.0000	0.0000	1.105	1s <sup>0.10</sup> 2p <sup>0.12</sup> 3d <sup>0.06</sup>
Va(2)	0.8610	0.6100	0.0700	0.801	1s <sup>0.06</sup> 2p <sup>0.04</sup> 3d <sup>0.01</sup>
Va(3)	0.3400	0.1050	0.1260	0.771	1s <sup>0.07</sup> 2p <sup>0.04</sup> 3d <sup>0.01</sup>
Va(4)	0.5000	0.5000	0.0000	0.821	1s <sup>0.06</sup> 2p <sup>0.04</sup> 3d <sup>0.01</sup>
Va(5)	0.5000	0.0000	0.0000	1.131	1s <sup>0.11</sup> 2p <sup>0.13</sup> 3d <sup>0.07</sup>
Va(6)	0.0000	0.5000	0.0000	1.141	1s <sup>0.09</sup> 2p <sup>0.11</sup> 3d <sup>0.06</sup>
<b>ReS<sub>2</sub></b>					
Re(1)	0.9432	0.690 35	0.5080	1.121	[Xe] 5d <sup>4.31</sup> 6s <sup>0.18</sup> 6p <sup>0.16</sup> 5f <sup>0.03</sup>
Re(2)	0.5119	0.7144	0.5020	1.141	[Xe] 5d <sup>4.40</sup> 6s <sup>0.21</sup> 6p <sup>0.19</sup> 5f <sup>0.03</sup>
S(1)	0.7486	0.6164	0.7858	1.695	[Ne] 3s <sup>1.92</sup> 3p <sup>4.63</sup> 3d <sup>0.42</sup>
S(2)	0.2758	0.6563	0.7574	1.695	[Ne] 3s <sup>1.91</sup> 3p <sup>4.65</sup> 3d <sup>0.38</sup>
S(3)	0.7723	0.8898	0.2761	1.695	[Ne] 3s <sup>1.91</sup> 3p <sup>4.71</sup> 3d <sup>0.34</sup>
S(4)	0.2462	0.8651	0.3023	1.695	[Ne] 3s <sup>1.91</sup> 3p <sup>4.77</sup> 3d <sup>0.31</sup>
Va(1)	0.0000	0.0000	0.0000	1.131	1s <sup>0.07</sup> 2p <sup>0.09</sup> 3d <sup>0.06</sup>
Va(2)	0.8800	0.6500	0.1150	0.763	1s <sup>0.07</sup> 2p <sup>0.04</sup> 3d <sup>0.01</sup>
Va(3)	0.3300	0.1000	0.0680	0.912	1s <sup>0.07</sup> 2p <sup>0.06</sup> 3d <sup>0.02</sup>
Va(4)	0.5000	0.5000	0.0000	1.022	1s <sup>0.08</sup> 2p <sup>0.08</sup> 3d <sup>0.04</sup>
Va(5)	0.5000	0.0000	0.0000	1.047	1s <sup>0.06</sup> 2p <sup>0.06</sup> 3d <sup>0.04</sup>
Va(6)	0.0000	0.5000	0.0000	1.022	1s <sup>0.07</sup> 2p <sup>0.07</sup> 3d <sup>0.03</sup>
<b>ReSe<sub>2</sub></b>					
Re(1)	0.936 85	0.685 23	0.508 62	1.224	[Xe] 5d <sup>4.58</sup> 6s <sup>0.28</sup> 6p <sup>0.23</sup> 5f <sup>0.03</sup>
Re(2)	0.487 23	0.290 42	0.496 81	1.224	[Xe] 5d <sup>4.58</sup> 6s <sup>0.29</sup> 6p <sup>0.24</sup> 5f <sup>0.03</sup>
Se(1)	0.751 84	0.618 08	0.793 86	1.720	[Ar] 4s <sup>1.91</sup> 4p <sup>4.02</sup> 4d <sup>0.49</sup> 4f <sup>0.24</sup>
Se(2)	0.720 49	0.339 02	0.238 68	1.693	[Ar] 4s <sup>1.90</sup> 4p <sup>4.02</sup> 4d <sup>0.43</sup> 4f <sup>0.19</sup>
Se(3)	0.226 10	0.108 62	0.723 66	1.769	[Ar] 4s <sup>1.92</sup> 4p <sup>4.16</sup> 4d <sup>0.49</sup> 4f <sup>0.24</sup>
Se(4)	0.245 04	0.863 54	0.303 36	1.845	[Ar] 4s <sup>1.94</sup> 4p <sup>4.39</sup> 4d <sup>0.54</sup> 4f <sup>0.28</sup>
Va(1)	0.835 66	0.151 18	0.058 66	0.968	1s <sup>0.07</sup> 2p <sup>0.06</sup> 3d <sup>0.02</sup>
Va(2)	0.000 00	0.000 00	0.000 00	1.137	1s <sup>0.07</sup> 2p <sup>0.07</sup> 3d <sup>0.04</sup>
Va(3)	0.500 00	0.500 00	0.000 00	1.077	1s <sup>0.08</sup> 2p <sup>0.08</sup> 3d <sup>0.04</sup>
Va(4)	0.500 00	0.000 00	0.000 00	1.034	1s <sup>0.06</sup> 2p <sup>0.05</sup> 3d <sup>0.02</sup>
Va(5)	0.000 00	0.500 00	0.000 00	1.050	1s <sup>0.07</sup> 2p <sup>0.06</sup> 3d <sup>0.03</sup>
Va(6)	0.396 56	0.680 00	0.081 64	0.789	1s <sup>0.06</sup> 2p <sup>0.04</sup> 3d <sup>0.01</sup>

of the wavefunctions of Tc 4d (Re 5d) orbitals owing to the shorter Tc–Tc (or Re–Re) distances.

The calculations show near the top of the valence band higher S 3p (Se 4p) DOSs for X(3) and X(4) than for X(1) and X(2), whereas at greater distance from the top the partial DOSs are higher for X(1) and X(2). This is a consequence of the changes in the chalcogen-to-metal distances  $X(4) > X(3) > X(2) > X(1)$ . At the bottom of the conduction band the Tc 4d (Re 5d) orbitals have a much higher DOS (about four times) than S 3p (Se 4p), indicating that the conduction band is mainly 4d or 5d of the transition metal.

The band-structure calculations performed for ReS<sub>2</sub> and ReSe<sub>2</sub> here are difficult to

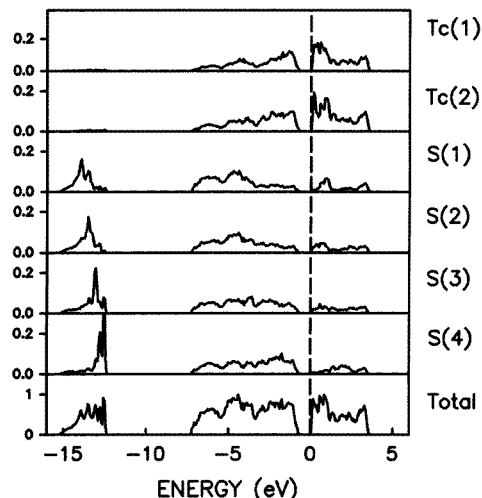


Figure 10. Partial and total DOSs for 1T-TcS<sub>2</sub>.

compare with the calculations by Whangbo and co-workers [5,9]. These workers present only DOS curves for a single slab in order to understand STM data. In addition their calculations are semiempirical.

#### 4. X-ray photoemission spectra

The crystals were grown by Wildervanck and Jellinek [2]. The crystals used were approximately squares with rounded sides, with dimensions of about 5–10 mm. X-ray photoelectron spectroscopy measurements were carried out in a small-spot ESCA machine from VG. A spot size of 300–600  $\mu\text{m}$  was used. The radiation source was an Al anode using the  $K\alpha$  line with a photon energy of 1486.6 eV. The sample surface was cleaned by stripping with Scotch tape in the preparation chamber at a base pressure of  $10^{-9}$  Torr. The sample with a fresh surface was transported to the main chamber (base pressure,  $10^{-10}$  Torr).

The wide range of x-ray photoelectron spectra of  $\text{ReSe}_2$  and  $\text{ReS}_2$  showed hardly any contamination by oxygen and carbon. The x-ray photoelectron spectra of the core levels are consistent with those in the literature for the elements [30] with consideration of chemical shifts, except for a higher-energy difference between  $\text{ReS}_2$  and  $\text{ReSe}_2$  due to the Fermi level difference ( $\text{ReS}_2$  is an n-type semiconductor; see below, where it is stated that  $\text{ReSe}_2$  grown with  $\text{I}_2$  as transport agent is a p-type semiconductor). Figure 11 shows the x-ray photoelectron spectra of the Re 4f core level electrons for  $\text{ReS}_2$  and  $\text{ReSe}_2$ . The peak splittings (Re  $4f_{7/2}$  and  $4d_{5/2}$ ) and peak widths are the same for  $\text{ReS}_2$  and  $\text{ReSe}_2$ ; the binding energies for  $\text{ReS}_2$  are shifted by about 1.0 eV to a higher energy owing to the Fermi level shift. This indicates that the Re atoms have the same valence in  $\text{ReSe}_2$  and  $\text{ReS}_2$ .

The x-ray photoelectron spectra of the valence bands for  $\text{ReS}_2$  and  $\text{ReSe}_2$  are shown in figure 12. The S 3s band of  $\text{ReS}_2$  has a width of about 5.0 eV (from  $-17.5$  to  $-12.5$  eV). The valence band of  $\text{ReS}_2$  (from  $-9.0$  to about  $-1.0$  eV) shows an energy gap of about 1 eV below the Fermi level, which means that  $\text{ReS}_2$  is an n-type semiconductor with a gap

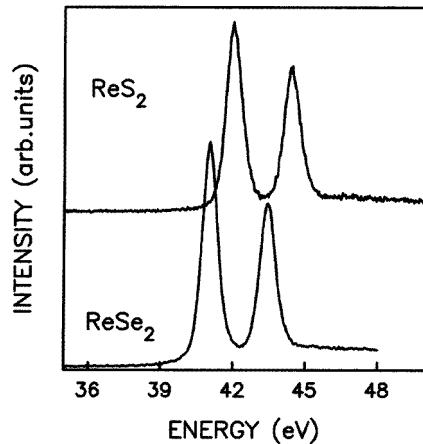


Figure 11. X-ray photoemission spectra of the 4f core levels of  $\text{ReS}_2$  and  $\text{ReSe}_2$ .

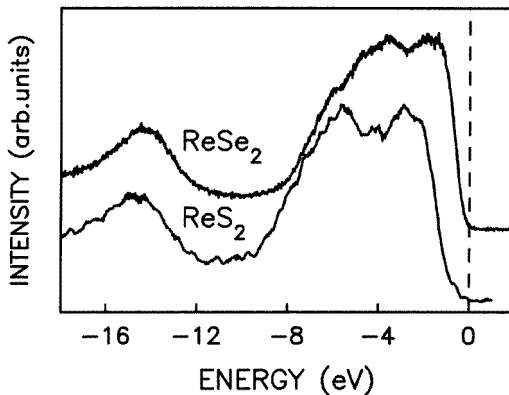
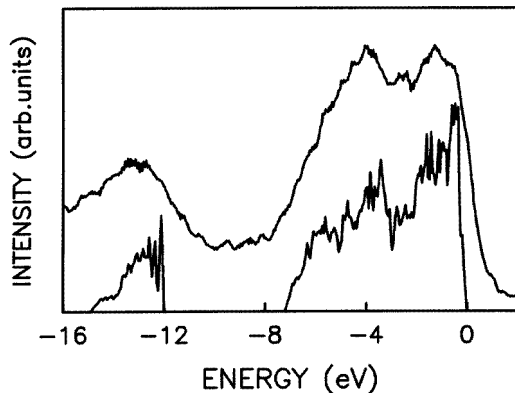


Figure 12. X-ray photoemission spectra of the valence bands of  $\text{ReS}_2$  and  $\text{ReSe}_2$ .

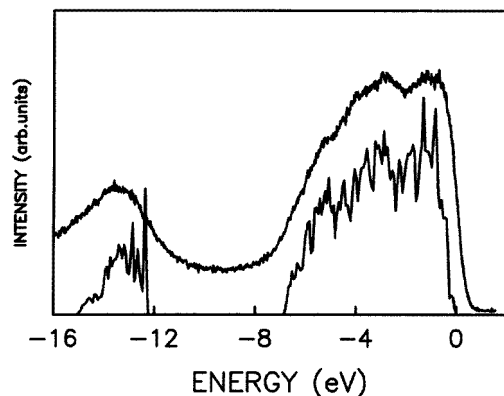
of about 1.0 eV, corresponding well to the energy gap of 1.32 eV deduced from the optical absorption spectra [2, 13–15]. The Se 4s band of  $\text{ReSe}_2$  ranges from  $-17.0$  to  $-12.0$  eV. The valence band begins from about  $-8.0$  eV to the Fermi level. The Fermi level is at the top of the valence band which confirms the p-type semiconductivity observed in transport measurements. Figure 13 and 14 show the x-ray photoelectron spectra of the valence band of  $\text{ReS}_2$  and  $\text{ReSe}_2$  and the spectrum calculated using the cross sections for photoemissions [31]. For comparison the spectra are shifted to a lower energy of about 1.6 eV and 0.6 eV for  $\text{ReS}_2$  and  $\text{ReSe}_2$ , respectively. If the spectra resolution (about 1.0 eV for photons with an energy of 1486.6 eV) is taken into consideration, the calculated spectra are in good agreement with the measurements.

## 5. Conclusions

Using the *ab-initio* LSW method the electronic structures for  $\text{ReS}_2$ ,  $\text{ReSe}_2$  and  $\text{TcS}_2$ , and their undistorted structures 3R- $\text{ReS}_2$ , 3R- $\text{ReSe}_2$  and 1T- $\text{TcS}_2$ , are reported. The undistorted



**Figure 13.** X-ray photoemission spectrum of the valence band of  $\text{ReS}_2$  and the calculated spectrum.



**Figure 14.** X-ray photoemission spectrum of the valence band of  $\text{ReSe}_2$  and the calculated spectrum.

structures have a trigonal-antiprismatic crystal-field splitting in the band structures. The Fermi level is in the  $d_{x^2-y^2}$  and  $d_{xy}$  part of the  $t_{2g}$  manifold and contains one electron; the  $d_{z^2}$  orbital is hybridized with chalcogen p and lies at a lower energy.

$\text{ReS}_2$ ,  $\text{ReSe}_2$  and  $\text{TcS}_2$  are semiconductors with energy gaps of about 1.0, 0.5 and 0.7 eV, respectively.

Photoelectron spectra for the core level and valence band of  $\text{ReS}_2$  and  $\text{ReSe}_2$  were obtained. The valence band spectra of  $\text{ReS}_2$  and  $\text{ReSe}_2$  are in good agreement with the band-structure calculations.

### Acknowledgment

We thank Dr H-J Lamfers for providing information on his crystal structure determination prior to publication and some figures.

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